Compliant thin film patterns of stiff materials as platforms for stretchable electronics

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A thin film of a stiff material, patterned as a serpentine on a flat elastomeric substrate, can elongate substantially when the substrate is pulled. We showed that the film elongates by twisting out of plane, accommodated by the compliance of the substrate and the pattern of the film. Consequently, large elongations of the substrate induce small strains in the film, even when the width of the film is much larger than its thickness. Such a wide serpentine, or other compliant patterns of stiff materials, can serve as a platform on which electronic circuits can be fabricated. This architecture will make electronics elastically stretchable.

A new direction for integrated circuit technology is to develop deformable electronic surfaces.^{1–6} Of the various modes of deformation (e.g., bending, twisting, and stretching), stretching is typically the most demanding, easily inducing a large tensile strain on the surface of a substrate. While an elastomeric substrate can recover from a large strain, most electronic materials, such as metals, dielectrics, and semiconductors, fracture at small strains (less than about one percent).^{7–11} How to use these materials to make stretchable electronic circuits remains uncertain. We propose that electronic circuits can be fabricated on a platform of a stiff material but a compliant pattern lying on an elastomeric substrate. These circuits will function without appreciable fatigue when the substrate is pulled for many cycles.

A helical spring can elongate substantially, even though the material of which it is made can sustain only a small strain. One could fabricate electronic circuits on a helical platform, but this approach would require microfabrication in three dimensions, a technology that requires substantial development itself. To be compatible with planar microfabrication technology, the platform must be planar. As an illustration of this principle, Fig. 1 shows a piece of paper cut into a serpentine and pulled at the two ends. While initially planar, the serpentine elongates by twisting out of plane so that a large elongation induces only small strains. The serpentine illustrates the principle that a film of a stiff material can be made compliant if the film is suitably patterned. It has been shown that serpentine metal interconnects on elastomeric substrates can sustain more than 200 cycles of elongation by 25%.¹² An existing design for stretchable electronics is to fabricate on a polymeric substrate small functional islands of stiff materials, which are then linked by metal interconnects.^{4,6,13} While the metal interconnects can be made stretchable,^{12,14} the crossovers of the interconnects from the polymer to the islands are susceptible to fracture. In addition, the size of each individual island must be kept small to avoid cracking.^{4,6,13}

The metal interconnects used in Ref. 12 had cross sections of comparable width and thickness. However, as demonstrated by the serpentine of paper, a large elongation can be achieved for any material and for serpentines of any width-to-thickness ratio. We propose that a thin film of a stiff material (e.g., silicon dioxide, silicon nitride, or a metal) can be patterned as a wide serpentine on a flat elastomeric substrate and serve as a platform on which entire electronic circuits can be fabricated using the planar microfabrication technology. Such a structure can sustain many cycles of large elongation without fracturing the electronic materials. The serpentine can have a much larger surface area than the islands that have been demonstrated to date.^{4,6,13}

For a film on a substrate to elongate substantially by twisting out of plane, two conditions must be met: the substrate must be sufficiently compliant, and the film must be suitably patterned. If the substrate were stiff or the film were straight, the film would deform within the plane, so that a small elongation of the substrate would induce a significant strain in the film. We next quantify the compliance of the substrate and the pattern of the film needed to achieve a large elongation.

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FIG. 1. (a) A piece of paper is cut into a serpentine. (b) When pulled, the serpentine elongates by twisting out of plane. (The reader may wish to try this experiment to see the serpentine twist in three dimensions.)

Figure 2 illustrates a model, which we analyze using the finite element code ABAQUS. The film is of a sinusoidal shape, period L, amplitude A, width W, and thickness h_{film} . The substrate is a $L \times L \times h_{\text{sub}}$ block, with displacement *u* prescribed on the two $L \times h_{sub}$ end surfaces. Marked in the figure is a point P, where the film reaches the maximum strain ϵ_{max} . To avoid confusion with the strain in the film, we call the quantity 2u/L the relative elongation of the substrate. The film is meshed with four-node quadrilateral shell elements, with 10 layers of elements along the width; all elements are nearly square. The substrate is meshed with eight-node linear brick elements, with size-matching elements at the film/ substrate interface and coarser elements far away from the interface. We model both the film and the substrate as linear elastic materials, with Young's modulus $E_{\rm film} =$ 100 GPa and $E_{sub} = 1$ to 100 MPa, so that E_{sub}/E_{film} ranges from 10^{-5} to 10^{-3} . The linear elastic assumption is reasonable for elastomeric substrates within strains of



FIG. 2. Schematic illustration of a thin film patterned in a sinusoidal shape on a substrate, which is then subject to elongation. The thickness of the film h_{film} (not shown in the figure) is much smaller than the width of the film *W*.

tens of percent. The assumption is also reasonable for the films, as strains are small in the films suitably patterned on compliant substrates. Poisson's ratio is taken to be 0.3 for both materials.

Figure 3 shows the deformed films bonded to substrates of various Young's moduli, subject to a relative elongation of 25%. For visual clarity, the figure does not show the substrates. The state of strain at each point in the film has two principal components, the larger of which is indicated by the shade in the figure. On a very compliant substrate, for which $E_{\rm sub}/E_{\rm film} = 10^{-5}$, the out-of-plane displacement of the surface is antisymmetric with respect to the x_2 axis; that is, to one side of the crest, the serpentine pushes the substrate down, and to the other side of the crest the serpentine lifts the substrate up. On a less compliant substrate, $E_{sub}/E_{film} =$ 10^{-3} , the out-of-plane displacement is symmetric with respect to the x_2 axis. As the modulus of the substrate increases, the displacement is gradually confined in the plane, and the strains in the film also increase. For a serpentine with a large width-to-thickness ratio, bending and stretching within the plane leads to a much larger strain than bending and twisting out of the plane.

Figure 4 plots the maximum strain in the film ϵ_{max} as a function of the relative elongation of the substrate 2u/L. For a film on a very compliant substrate (e.g., $E_{\text{sub}}/E_{\text{film}} = 10^{-5}$), $\epsilon_{\text{max}} < 3.5\%$ at a relative elongation of 25%. When the modulus of the substrate increases, so does ϵ_{max} . For example, when $E_{\text{sub}}/E_{\text{film}} = 10^{-3}$, $\epsilon_{\text{max}} =$ 11.6% at the relative elongation of 25%. Figure 4 also shows that ϵ_{max} increases as the substrate becomes thicker. Further calculations show, however, that ϵ_{max}



FIG. 3. Deformed films on substrates of various Young's moduli, subject to a relative elongation of 25%. For visual clarity, the substrates are not shown. Here W/L = 0.05, A/L = 0.5, $h_{\rm film}/L = 0.005$, $h_{\rm sub}/h_{\rm film} = 10$. The shades indicate the levels of the larger principal strain in the film. The three figures are viewed approximately at angles 0°, 15°, and -10° from the direction normal to the substrate surface.



FIG. 4. Maximum strain in the film ϵ_{max} as a function of the relative elongation of the substrate. Here W/L = 0.05, A/L = 0.5, $h_{film}/L = 0.005$.

becomes insensitive to the thickness of the substrate when h_{sub}/L exceeds unity. This limiting case is shown by the curves for $h_{sub}/L = 5$.

When the substrate is sufficiently compliant, ϵ_{max} is also insensitive to the width and the thickness of the serpentine, and is only sensitive to its amplitude-toperiod ratio *A/L*. Figure 5 shows ϵ_{max} as a function of *A/L*. Here we simulate a freestanding film, corresponding to the limiting case of a film on an infinitely compliant substrate. If the film is a straight stripe (*A/L* = 0), ϵ_{max} is identical to the relative elongation. If *A/L* > 0, the patterned film can benefit from both in-plane bending and out-of-plane twisting. The larger the value of *A/L*, the smaller the strains level in the film. When *A/L* > 1, ϵ_{max} is more than twenty times smaller than that of a straight stripe.

Figure 6 plots the maximum in-plane strain in the substrate and the maximum interfacial stresses as functions



FIG. 5. Effect of the sinusoidal pattern on the maximum strain in the film $\varepsilon_{\rm max}$



FIG. 6. The maximum elongation in the substrate and the maximum interface stresses as a function of the relative elongation of the substrate.

of the relative elongation of the substrate. The maximum in-plane strain in the substrate occurs on its surface near the arms of the film serpentine, with a magnitude slightly higher than that of the applied relative elongation. The strains in the substrate become uniform about one serpentine period below the surface. As the relative elongation of the substrate increases, the maximum normal stress on the film/substrate interface increases substantially when the film deflects out of plane, and then saturates. The maximum shear stress on the interface increases nearly linearly. The more compliant the substrate, the smaller the maximum interfacial stresses. As shown in Fig. 6, the maximum interfacial stress is below 1 MPa when a very compliant substrate is used.

As the elongation increases, the serpentine will be finally straightened, as shown in Fig. 1(b). Further elongation leads to a pure stretch of the film. Let $L_{\rm arc}$ be the arc length of a serpentine of period *L*. The relative elongation upon which a serpentine is fully straightened can be estimated by $(L_{\rm arc} - L)/L$. For the sinusoidal stripe, the maximum relative elongation is given by

$$\frac{L_{\rm arc} - L}{L} = \frac{1}{\pi} \int_0^{\pi} \left[1 + \left(\frac{A}{L}\right)^2 \pi^2 \sin^2 x \right]^{1/2} dx - 1 \quad .$$
(1)

This equation is plotted in Fig. 7, which provides an estimate of the A/L ratio needed to achieve a certain relative elongation.

Most electronic materials fracture at strains less than about 1%. To limit strains in a serpentine below 1%, the relative elongation should be kept below 20% for a serpentine of A/L = 1 on a very compliant substrate (Fig. 5). Larger relative elongation can be achieved by optimizing the pattern of the film, for example, by increasing the amplitude of the stripe and decreasing its



FIG. 7. Maximum relative elongation upon which a sinusoid is fully straightened as a function of A/L.

width at crests and troughs. Also, the out-of-plane twisting of a serpentine requires only the compliance of a top layer of the substrate, of a thickness that scales with the feature size of the film pattern, such as the period L and the amplitude A. Underneath such a compliant top layer, a less compliant material may be used as a backing if some overall rigidity is desired.

We have focused on a serpentine on an elastomeric substrate subject to uniaxial elongation. With suitable choices of length ratios, such a structure can also sustain biaxial elongation. Furthermore, a network of serpentines not only can sustain biaxial elongation, but also has bidirectional connectivity. In fact, a large variety of patterns allows substantial elongation by the same principle. In a future study, we will explore such patterns to meet various design requirements.

In summary, we have shown that a thin film of a stiff material, suitably patterned on a sufficiently compliant substrate, elongates by twisting out of plane, so that large elongations of the substrate induce small strains in the film. We propose that such patterned films can serve as platforms on which entire electronic circuits can be fabricated using planar microfabrication technology. Such circuits will function without appreciable fatigue when the substrate is repeatedly bent, twisted, and stretched.

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REFERENCES

- J.A. Rogers and Z. Bao: Printed plastic electronics and paperlike displays. J. Polym. Sci., Part A: Polym. Chem. 40, 3327 (2002).
- S.R. Forrest: The path to ubiquitous and low-cost organic electronic appliances on plastic. *Nature* 428, 911 (2004).
- 3. G.P. Collins: Next stretch for plastics electronics. *Sci. Am.* **291**(2), 74 (2004).
- S. Wagner, S.P. Lacour, J. Jones, P.I. Hsu, J.C. Sturm, T. Li, and Z. Suo: Electronic skin: Architecture and components. *Physica E* 25, 326 (2005).
- T. Sekitani, Y. Kato, S. Iba, H. Shinaoka, T. Someya, T. Sakurai, and S. Takagi: Bending experiment on pentacene field-effect transistors on plastic films. *Appl. Phys. Lett.* 86, 073511 (2005).
- R. Bhattacharya, S. Wagner, Y-J. Tung, J. Esler, and M. Hack: OLED pixel array on a dome. *Proc. IEEE* 93, 1273 (2005).
- H. Gleskova, S. Wagner, and Z. Suo: Stability of amorphous silicon transistors under extreme in-plane strain. *Appl. Phys. Lett.* 75, 3011 (1999).
- H. Huang and F. Spaepen: Tensile testing of free-standing Cu, Ag and Al thin films and Ag/Cu multilayers. *Acta Mater.* 48, 3261 (2000).
- S.L. Chiu, J. Leu, and P.S. Ho: Fracture of metal-polymer line structures. I. Semiflexible polyimide. J. Appl. Phys. 76, 5136 (1994).
- B.E. Alaca, M.T.A. Saif, and H. Sehitoglu: On the interface debond at the edge of a thin film on a thick substrate. *Acta Mater.* 50, 1197 (2002).
- T. Li, Z.Y. Huang, Z. Suo, S.P. Lacour, and S. Wagner: Stretchability of thin metal films on elastomer substrates. *Appl. Phys. Lett.* 85, 3435 (2004).
- D.S. Gray, J. Tien, and C.S. Chen: High-conductivity elastomeric electronics. Adv. Mater. 16, 393 (2004).
- P.I. Hsu, M. Huang, Z. Xi, S. Wagner, Z. Suo, and J.C. Sturm: Spherical deformation of compliant substrates with semiconductor device islands. *J. Appl. Phys.* **95**, 705 (2004).
- S.P. Lacour, S. Wagner, Z.Y. Huang, and Z. Suo: Stretchable gold conductors on elastomeric substrates. *Appl. Phys. Lett.* 82, 2404 (2003).